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Journal of  
The Maharaja Sayajirao University of Baroda

## Certificate of Publication

Certificate of publication for the article titled:

**INFLUENCE OF Ni<sup>2+</sup> IONS ON THE STRUCTURAL AND WETTABILITY PROPERTIES  
OF Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> SPINEL FERRITE THIN FILMS DEPOSITED BY SPRAY PYROLYSIS  
METHOD**

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Volume No .56 No.1(VIII) 2022

Approved in Journal

Journal of The Maharaja Sayajirao University of Baroda

ISSN : 0025-0422

(UGC CARE Group I Journal)



Journal MSU of Baroda

**INFLUENCE OF Ni<sup>2+</sup> IONS ON THE STRUCTURAL AND WETTABILITY PROPERTIES  
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**Abstract:**

In the current study, cobalt nickel ferrite (Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub>) is used. The structural, optical, and wettability features of the produced cobalt ferrite thin film were studied and optimized before being placed on glass substrate by spray pyrolysis, investigations with the aid of UV-VIS spectroscopy and X-ray diffraction (XRD). At room temperature, the XRD pattern was measured in the 2θ range of 20 to 80 degree. The cubic spinel structure is represented by all of the reflections in the XRD pattern. The structural characteristics that were computed from the XRD data, such as the lattice parameter and X-ray density, matched the conventional JCPDS data quite well. The generated cobalt ferrite thin film's nanocrystalline nature was confirmed by the crystallite size, which Scherer's formula determined to be 28 nm. The near edge band emission was visible in the PL spectra between 690 and 740 nm in wavelength. With increasing the Ni<sup>2+</sup> content x, it was discovered that the optical band gap computed using the Tauc plot was in the range of 2.43 eV to 2.61 eV. It was discovered that the contact angle ranged from 55.99° to 87.50°, indicating the hydrophilic nature of all the samples.

**INTRODUCTION:**

In recent years, a nanocrystalline spinel cobalt ferrite thin film has been used for gassensing and supercapacitors applications. The crystal structure of cobalt ferrite possesses inverse spinel structure with one-half of Fe<sup>3+</sup> ions on (A) site and rest together with Co<sup>2+</sup> ions at [B] site at room temperature. A variety of physical as well as chemical methods have been applied for the deposition of cobalt ferrite thin films [5,9-11]. On the other hand, Nickel ferrite (NiFe<sub>2</sub>O<sub>4</sub>) is one of the versatile and technologically important soft ferrite (low anisotropy field) materials with spinel structure because of its typical ferromagnetic properties, low electrical conductivity and thus lower eddy current losses, high electrochemical stability, catalytic behavior and abundance in nature [12, 13]. Cobalt ferrite is a well-known hard magnetic material with relatively high coercivity and saturation magnetization while nickel ferrite is a soft material with low coercivity and saturation magnetization. Many of these (hard and soft magnetic) properties make them very promising candidates for a variety of applications in biomedical, electronic as well recording technologies [14-16]. From the application point of view, the magnetic character of the nanoparticles depends crucially on the size, shape, purity, and magnetic stability of these nanoparticles. CoFe<sub>2</sub>O<sub>4</sub> and NiFe<sub>2</sub>O<sub>4</sub> are room-temperature insulating ferrimagnetism with high

Curie temperature and large saturation magnetization make them promising candidates for applications in spin-filter devices or as building blocks of artificial multiferroic heterostructures. For all these possible applications the spinel ferrites have to be grown on suitable substrates, which usually incorporate strain into the thin film material. This strain can have a strong influence on the structural and magnetic properties of the grown structures. It has been shown experimentally for CoFe<sub>2</sub>O<sub>4</sub> that, depending on growth conditions and substrate treatment. Furthermore, a strong enhancement of magnetization and conductivity has been reported in NiFe<sub>2</sub>O<sub>4</sub>. From the experimental point of view, it is not clear whether the observed deviations from bulk behavior are determined solely by the strain incorporated in the thin films.

Physical methods include pulsed laser deposition, sputtering, etc. Physical methods have many drawbacks such as small area of deposition, the requirement of sophisticated instruments, high working cost of the system, wastage of depositing material, cleaning after each deposition, etc. Keeping drawbacks of physical methods in mind, recently, several chemical methods are used for the preparation of ferrite thin films. On the other hand, chemical methods are simple, economical, and

convenient for the deposition of metallic thin films. The different preparative parameters are easily controllable. Chemical methods including sol-gel route, low-pressure chemical vapor deposition, electrodeposition, and so on are considered for the synthesis of spinel ferrite thin films of fascinating magnetic properties. Out of all these chemical methods, the spray pyrolysis method is the most convenient and suitable for the preparation of spinel ferrite thin films. This can be done either by varying the sizes of these nanoparticles or by adjusting the concentrations of soft (e.g. nickel ferrite) and hard (e.g. cobalt ferrite) magnetic phases in these materials. For this purpose nickel substituted cobalt ferrite nanoparticles have been synthesized by spray pyrolysis technique. In the present work, we report the structural and wettability properties of Ni<sup>2+</sup> substituted CoFe<sub>2</sub>O<sub>4</sub> thin films in detail.

### Experimental

Analytical reagent (AR) grade nickel nitrate (Ni (NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O), cobalt nitrate (Co(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O), and ferric nitrate (Fe (NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O) were used as starting material without any purification. The glass substrates were carefully cleaned by sequential treatment with chromic acid, ethanol, and acetone followed by dipping with deionized water in an ultrasonic bath for 30 min, and the same glass plates were used for the deposition. The glass substrate cleaning plays an important role in the deposition of thin films. The extreme cleanliness of the glass substrate is required for the chemical deposition. The contaminated surface provides nucleation sites facilitating growth resulting in non-homogeneous films with different orientations and impurities. The glass microslides supplied by 'Blue Star' of dimension 75 x 25 x 1.45 mm have been used as the substrates for the deposition of cobalt-nickel ferrite thin films. The glass slides were washed with chromic acid and distilled water. The substrates were washed with double distilled water. The substrates were ultrasonically cleaned for 30 minutes by using an ultrasonic bath. Finally, the substrates were dried, degreased in AR-grade acetone, and were used for deposition. The Ni<sup>2+</sup> substituted CoFe<sub>2</sub>O<sub>4</sub> thin film was prepared by spray pyrolysis technique. Analytical reagent grade chemicals Co (NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O and Fe (NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O were used as starting materials. The two metal nitrate solutions were dissolved separately in deionized water at the concentration of 0.1 M for Co<sup>2+</sup>, Fe<sup>3+</sup>, and Ni<sup>2+</sup> solutions. Final solutions were prepared by mixing these two solutions in 1:2 volumetric proportions. Ni<sup>2+</sup> substitution CoFe<sub>2</sub>O<sub>4</sub> thin films were prepared by spraying the solution onto a previously cleaned glass substrate. Glass substrates mounted on a holder were placed on the surface of a hot plate. A temperature controller was used to hold the preset temperature of 350<sup>o</sup>C with an accuracy of ± 500<sup>o</sup>C through a channel alumel thermocouple connected to the glass substrate. A prepared solution was atomized in the air via a pneumatic spray system under an air pressure of 0.25 MPa. The atomized droplets were transformed onto the heated glass substrate for 0.5 sec intermittently. The substrate temperature could be reduced under the effect of spray and requires several seconds to recover the preset temperature. The solution spray rate was 5 ml / min, and the distance between nozzle 28 cm and substrate temperature 350<sup>o</sup>C were kept constant. When an aqueous solution of cobalt, nickel, and ferric nitrate is sprayed over the hot substrates, fine droplets of solution thermally decompose after falling over the hot surface of substrates, resulting in the formation of well adherent and homogeneous Ni<sup>2+</sup> substitution CoFe<sub>2</sub>O<sub>4</sub> thin films. The prepared thin films were annealed at 500<sup>o</sup>C for 4 h.

### RESULT AND DISCUSSION:

#### X-ray diffraction

The X-ray diffraction patterns of Ni<sup>2+</sup> substitution CoFe<sub>2</sub>O<sub>4</sub> thin films annealed at 500<sup>o</sup>C for 4 h show the crystallinity of the final product as given in **Fig. 1**. All the peaks in the XRD pattern correspond to the cubic CoFe<sub>2</sub>O<sub>4</sub> and NiFe<sub>2</sub>O<sub>4</sub> phases according to the standard JCPDS cards (742081 for NiFe<sub>2</sub>O<sub>4</sub> and 791744 for CoFe<sub>2</sub>O<sub>4</sub>). The **Fig. 1** shows XRD patterns of nanocrystalline Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x=0.0, 0.2, 0.4, 0.6, 0.8 and 1.0) thin films. The reflections (111), (220), (311), (222), (400), (422) (511), and (440) belonging to single phase cubic spinel structure, are present in the XRD pattern. The lattice constant was calculated using the interplanar spacing distance (d) and the respective (hkl) parameters using the following relation.

$$a = \frac{\lambda [h^2 + k^2 + l^2]^{1/2}}{2 \sin \theta} A^0 \text{ ----- (1)}$$

All prepared samples could be refined using the Fd3m space group and have a single-phase spinel cubic structure. The lattice parameters of prepared ferrite samples are listed in **Table:1**. The values of the lattice parameters of Ni<sup>2+</sup> substitution CoFe<sub>2</sub>O<sub>4</sub> are in agreement with that reported by Ashok

Kumar *et al.*, [17]. It has been found that the lattice parameter decreases from 8.382 Å to 8.332 Å with the increase in Ni<sup>2+</sup> concentration. The decrease in lattice parameter with the increase in Ni<sup>2+</sup> ion concentration is due to the replacement of large ionic radii Co<sup>2+</sup> ions (0.78 Å) by smaller ionic radii Ni<sup>2+</sup> ions (0.63 Å) in the host lattice of cobalt ferrite thin film.

The X-ray density of all the thin film samples was calculated using the volume of the unit cell and molecular weight using the following formula.

$$d_x = \frac{8M}{N_A a^3} (\text{gm/cm}^3) \dots (2)$$

where  $M$  is molecular weight,  $N$  is Avogadro's number, and ' $a$ ' is the lattice parameter. The values of X-ray density with Ni<sup>2+</sup> content are presented in **Table :1**. The X-ray density is found to increase with increasing Ni<sup>2+</sup> content  $x$ . As the volume of the thin films on the substitution of Ni<sup>2+</sup> content  $x$  increases the X-ray density decreases. The crystallite size of the deposited thin films has been estimated from the most intense peak (311). The crystallite size calculated for all the thin films as a function of Ni<sup>2+</sup> content  $x$  using Debye-Scherrer's formula is tabulated in **Table :1**

$$t = \frac{0.94\lambda}{\beta \cos \theta} \quad (\text{nm}) \dots (3)$$

where  $\lambda$  is the wavelength of the X-ray radiation,  $\beta$  is the full-width half maximum and  $2\theta$  is the diffraction angle. The crystallite size of Cu<sub>1-x</sub>Zn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> was found to be in the range of 17- 33 nm. Thus, the nanocrystalline nature of the deposited Ni<sub>2+</sub> substitution cobalt ferrite thin films was confirmed.

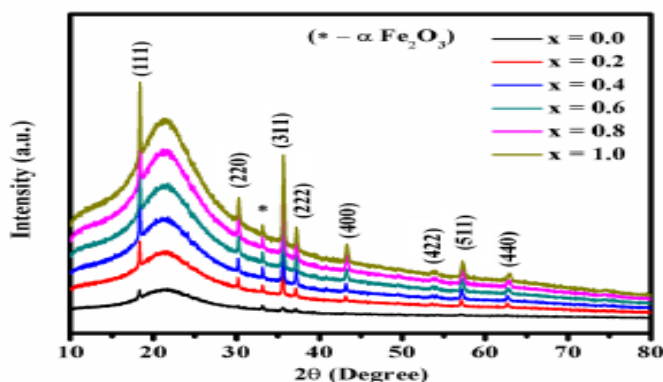


Figure 1. X- ray diffraction patterns of Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> thin films

Lattice parameter (a), X-ray density and Crystallite size (t) of Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0, 0.2, 0.4, 0.6, 0.8 and 1.0) thin films

Composition 'x'	a (Å)	d <sub>x</sub> (gm/cm <sup>3</sup> )	(t) (nm)
0.00	8.382	5.293	17
0.20	8.379	5.312	21
0.40	8.372	5.352	23
0.60	8.365	5.355	27
0.80	8.351	5.357	31
1.00	8.332	5.381	33

### Wettability Properties

The surface wettability of many nanomaterials is almost driven by the tendency to reduce the total Gibbs free energy of the structure via reducing the surface energy. It depends on the composition, morphology, surface roughness, local homogeneity, etc. The aperture on the film surface could induce

the air pocket forming and it will let the water drop and easily stand on the surface of as-deposited ferrite thin films. The aperture size and the roughness effect of Ni<sub>2+</sub> substitution cobalt ferrite thin films without and with a post-annealing process. The topography of the thin films was further supported by the contact angle measurement. The wettability of Ni<sub>2+</sub> substitution cobalt spinel ferrite thin films was evaluated by measuring the contact angle of a water droplet placed on the surface of the thin film using a contact angle meter. The contact angle recorded for Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0, 0.2, 0.4, 0.6, 0.8 and 1.0) is in the range of 55.99° - 87.50° which indicates the hydrophilic nature in **Fig :2**. The values of contact angle for each thin film are given in **Table :2**.

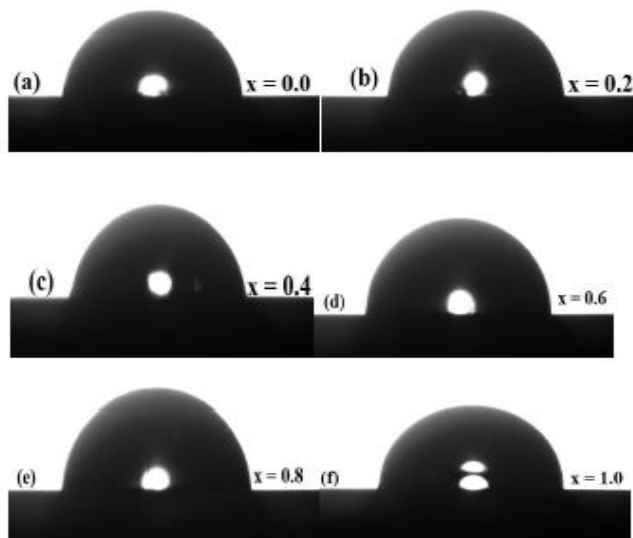


Figure 2 Contact angle of Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x=0.0, 0.2, 0.4, 0.6, 0.8 and 1.0) thin films.

Contact angle (C.A.), surface energy (S.E.), Energy Band gap (E<sub>g</sub>) and thickness of Co<sub>1-x</sub>Ni<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0, 0.2, 0.4, 0.6, 0.8 and 1.0) thin films

Composition 'x'	C. A. θ	S.E. (m J/m <sup>2</sup> )	E <sub>g</sub> (eV)	Thickness of thin film (nm)
0.0	65.47	44.49	2.61	249
0.2	55.99	50.24	2.60	253
0.4	84.28	32.80	2.57	274
0.6	80.90	34.92	2.51	254
0.8	88.28	30.46	2.49	265
1.0	87.50	30.50	2.43	267

## CONCLUSION

The nickel substituted cobalt ferrite thin films of various compositions were successfully grown on preheated glass substrates using the spray pyrolysis technique. All the films show uniform thickness ranging between 249 nm - 274 nm. Structural studies carried out by the X-ray diffraction method confirm the formation of a single-phase cubic spinel structure. Lattice constant was found to be decreased with an increase in nickel content x. Crystallite size was found to be increased with an increase in nickel content x. The contact angle was found to be in the range of 55.99° - 87.50° which indicates the hydrophilic nature of all the samples.

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